

## **THERMAL STABILITY AND STRUCTURAL STUDIES IN THE TeO<sub>2</sub>-ZnO-MgO-Li<sub>2</sub>O-Er<sub>2</sub>O<sub>3</sub> GLASS SYSTEM**

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### **ABSTRACT**

Series of (80-x)TeO<sub>2</sub>-18ZnO-1MgO-1Li<sub>2</sub>O-xEr<sub>2</sub>O<sub>3</sub> glass system (0.5mol%≤x≤2.5mol%) has successfully been made by melt quenching technique. The thermal stability and structure of glass has been investigated by means of TG/DTA and FTIR spectroscopy. The thermal parameters, such as the glass transition temperature (T<sub>g</sub>), crystallization temperature (T<sub>c</sub>) and thermal stability (T<sub>c</sub>-T<sub>g</sub>) were determined. It is found that this system provides a wide and stable glass formation in which the glass stability around 97°C-117°C may be obtained. The broad absorption peaks were observed around 657cm<sup>-1</sup>-671cm<sup>-1</sup> and 755cm<sup>-1</sup>-758cm<sup>-1</sup>, which correspond to the stretching vibration mode of TeO<sub>4</sub> tpb and TeO<sub>3</sub> tp, respectively. The absorption peaks around 1600cm<sup>-1</sup> and 3400cm<sup>-1</sup> are assigned to a stretching vibration of the hydroxyl group participating in the strong metal and hydrogen bonding respectively.

### **INTRODUCTION**

Tellurium dioxide (TeO<sub>2</sub>) is the most stable oxide of tellurium (Te), with a melting point of 773°C. The stability of tellurium oxide is one of the properties that originally attracted researchers, first to the crystalline solids and then to tellurite glasses [1]. Tellurite glasses are known to be an important amorphous system that has many potential commercial applications. The TeO<sub>2</sub>-ZnO glass system is expected to have a unique optoelectronic properties [2] because of not only their low transition temperature but also their excellent infrared transmission [3] in the range of 0.4 – 6.0μm [4], which give them potential application in pressure sensors or a new laser host [1] and thermally stable for fiber drawing [5]. Zinc tellurite glasses are reported to be suitable host for optically active rare earth ions [6]. It has also been reported elsewhere [7] that emission results of newly fabricated double-clad Er<sup>3+</sup>-doped tellurite single mode fibers, showing their potential use for fiber lasers and optical amplifier. Er<sup>3+</sup>-doped tellurite glasses have been attractive research subjects for up-conversion emission due to their low phonon energy [5].

To produce the better laser glass, it is necessary to know the thermal behavior, structural changes and phase existence, which might occur during the glass formation. Almost all studies of the TeO<sub>2</sub>-ZnO glass system are concerned with glass formation and structural property [2,4]. From FTIR spectral, a linear evolution of the glass network with composition may be seen. To develop such glass, a study on their structural properties is very important. Structural measurements in tellurite glasses are very essential to interpret their physical and chemical properties. The stability of

tellurite glasses lends itself to the need for reliable structural measurements [1]. This paper will report the latest development of structural properties on the glass. All the results will be discussed with respect to their composition.

### **EXPERIMENTAL DETAILS**

The Erbium doped zinc-tellurite glasses in the  $(80-x)\text{TeO}_2-18\text{ZnO}-1\text{MgO}-1\text{Li}_2\text{O}-x\text{Er}_2\text{O}_3$  system ( $0.5\text{mol}\% \leq x \leq 2.5\text{mol}\%$ ) were prepared by melt quenching technique. Batches of 10 g were prepared from commercial powders of  $\text{TeO}_2$ ,  $\text{ZnO}$ ,  $\text{MgO}$ ,  $\text{Li}_2\text{O}$  and  $\text{Er}_2\text{O}_3$ . A well-mixed mixture was milled for 0.5h before being melted in silica crucible at  $900^\circ\text{C}$  for 0.5h. After a required viscosity is achieved, the melts is quenched between two brass plates followed by annealing at  $250^\circ\text{C}$  for 5h before allowed to cool down to room temperature.

Pyris Diamond TG/DTA (Thermogravimetri / Differential Thermal Analyzer) was used to determine the thermal characteristics of the glasses. 10 mg glass samples were heated in the TG/DTA furnace with a heating rate of  $10^\circ\text{C}/\text{min}$  from  $25^\circ\text{C}$  to  $1000^\circ\text{C}$ . The Parkin Elmer GX FT-IR spectroscopy has been used to get the IR spectra. Typically about 2 mg of a finely ground sample is mixed with 200 mg of KBr before being pressed in to a pellet. The IR spectra were recorded in the range of  $400-4000\text{ cm}^{-1}$ .

### **RESULTS AND DISCUSSION**

Series of  $(80-x)\text{TeO}_2-18\text{ZnO}-1\text{MgO}-1\text{Li}_2\text{O}-x\text{Er}_2\text{O}_3$  glass system in the range ( $0.5\text{mol}\% \leq x \leq 2.5\text{mol}\%$ ) have successfully been made and their corresponding thermal parameter are summarized in the Table 1. Meanwhile, the TG/DTA curves from some sample are shown in Fig.1. The DTA curve for the glasses show broad endothermic hump corresponding to the glass transition temperature  $T_g$ . This transition is followed by one exothermic peak corresponding to crystallization temperature  $T_c$  and other endothermic event corresponding to the melting temperature  $T_m$ . The difference temperature  $\Delta T (=T_c - T_g)$  corresponding to the thermal stability of glasses in also calculated. The relationship between  $T_g$ ,  $T_c$  and  $T_c - T_g$  as a function of mol%  $\text{Er}_2\text{O}_3$  was plotted and shown in Fig 2.

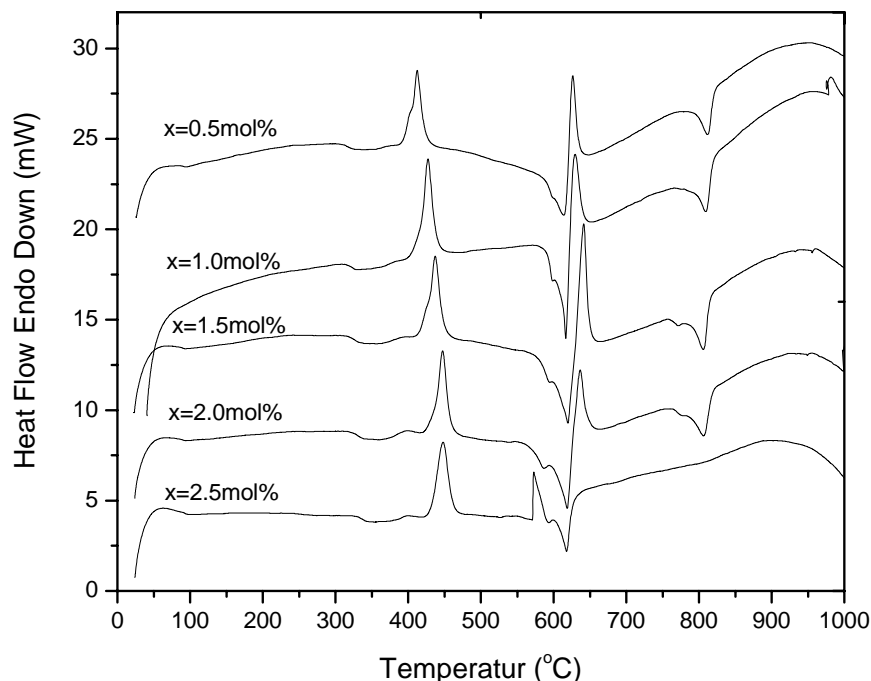


Figure 1: DTA curve of  $(80-x)\text{TeO}_2-18\text{ZnO}-1\text{MgO}-1\text{Li}_2\text{O}-x\text{Er}_2\text{O}_3$

Tabel 1: Glass composition and thermal characteristics

Sample No	Composition (mol %)					Temperature (°C)			
	TeO <sub>2</sub>	ZnO	MgO	Li <sub>2</sub> O	Er <sub>2</sub> O <sub>3</sub>	T <sub>g</sub>	T <sub>c</sub>	T <sub>c</sub> -T <sub>g</sub>	T <sub>m</sub>
C1	79.5	18.0	1.0	1.0	0.5	315	412	97	615
C2	79.0	18.0	1.0	1.0	1.0	321	427	106	617
C3	78.5	18.0	1.0	1.0	1.5	322	437	115	620
C4	78.0	18.0	1.0	1.0	2.0	330	447	117	619
C5	77.5	18.0	1.0	1.0	2.5	332	448	116	618

From Table 1 and Fig 2, as the addition doping of Er<sub>2</sub>O<sub>3</sub> content in the tellurite glasses from 0.5mol% to 2.5mol% results in regular increases of T<sub>g</sub> from 315°C to 332°C. The glass transition temperature increases with addition of Er<sub>2</sub>O<sub>3</sub> in the glass system due to the cleavage of the networks implies increases in the rigidity of the network formed by TeO<sub>3</sub> tp units and the increase of NBO atoms in the glasses [8,9]. Thus, the increased Er<sub>2</sub>O<sub>3</sub> content, the crystallization temperature from 412°C to 448°C (T<sub>c</sub>) and increasing the thermal stability from 97°C to 117°C affect the ease of crystallization [10]. The increase in glass stability is also reported to be due to the structural formation of ZnTeO<sub>3</sub> units [11].

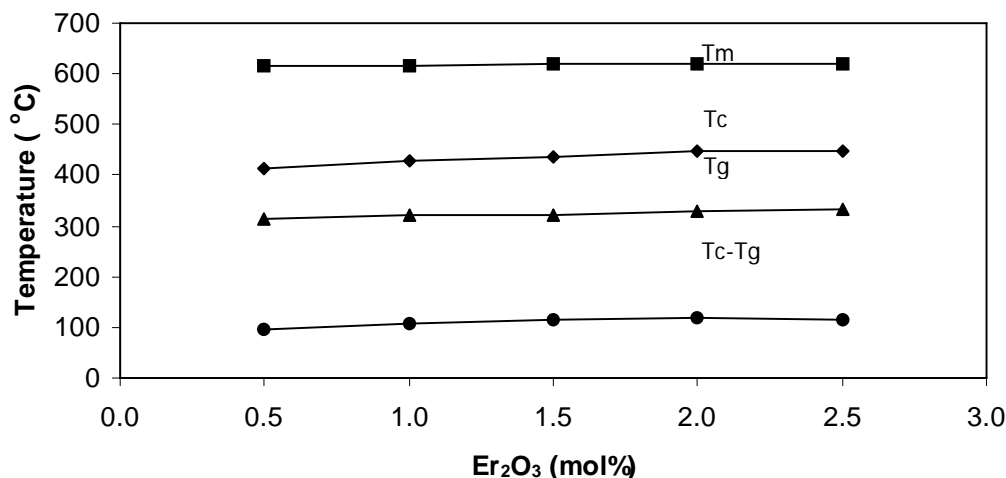


Figure 2: The relationship between  $T_g$ ,  $T_c$ ,  $T_c - T_g$  as a function of  $Er_2O_3$  content.

The melting temperature  $T_m$  increases from 615 °C to 620°C when  $Er_2O_3$  dopant from 0.5mol% to 1.5mol% respectively, indicating that the rigidity of the network the glass increase [9].

The IR transmission spectra of glasses are given in Figure 3 and their corresponding absorption bands are summarized in the Table 2. As can be observed from Figure 3 and Table 2, the absorption peaks are consistently observed at  $458cm^{-1}$  with the increase of the  $Er_2O_3$  content and this can be assigned to the Zn–O tetrahedral bond [12]. It should however be noted that there are many small peaks occur within these ranges as the  $Er_2O_3$  content is being increased. The occurrence of these small peaks is presumably due to the deformation of the Te–O bond vibration [4,13].

The broad peaks that were observed around  $650cm^{-1}$  and  $770cm^{-1}$  can be ascribed to the mixing structures of  $TeO_3$  groups, symmetric  $TeO_4$  groups and deformed  $TeO_4$  groups respectively [14]. However, as the  $Er_2O_3$  content increases, the sharp absorption peaks shift from  $657cm^{-1}$  to  $671cm^{-1}$ . This may be due to the mixing structure of symmetric and deformed  $TeO_4$  groups in this glass of  $Zn_2Te_3O_8$  [4]. For  $Er_2O_3$  content,  $0.5mol\% \leq x \leq 2.5mol\%$ , the absorption peak is slightly blue shift [14]. At the same time, the shoulder at  $758cm^{-1}$  and  $755cm^{-1}$  started to emerge. This can be assigned to the  $TeO_3$  tp of  $ZnTeO_3$  [3]. This might be due to the perturbation of  $TeO_4$  tbp unit into  $TeO_3$  tp unit via the intermediate coordination of  $TeO_{3+1}$  [15,16,17]. Sekiya et al. reported that  $TeO_{3+1}$  polyhedra and  $TeO_3$  tp units including NBO ions were formed in glasses with addition of the alkali oxides [18].

Mean while, the absorption peaks around  $1110cm^{-1}$ , which corresponds to the Te–O–Zn linkages [12], seem do not affected by the variation in composition.

Another well known IR absorption peaks occur around  $1600cm^{-1}$  and  $3400cm^{-1}$ , these are assigned to a stretching vibration of the hydroxyl group participating in the strong metal bonding as well as in the hydrogen bonding respectively [13]. The existence of these groups is very common to the oxide glass system.

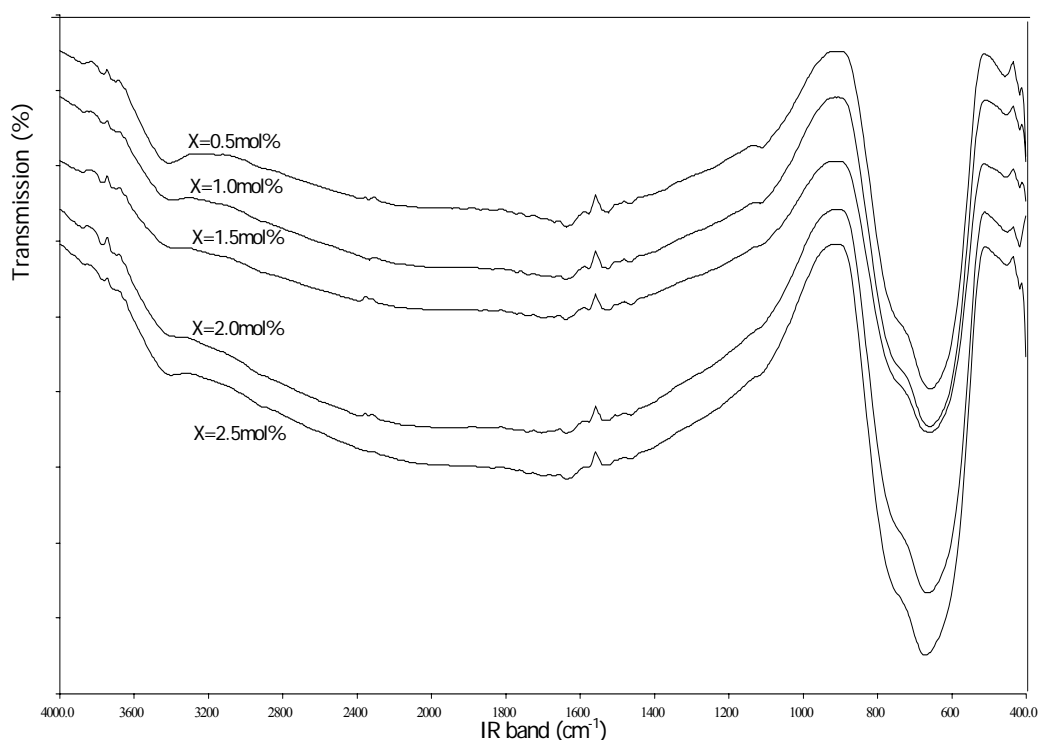


Figure 3: Infrared transmission spectra of  $(80-x)\text{TeO}_2-18\text{ZnO}-1\text{MgO}-1\text{Li}_2\text{O}-x\text{Er}_2\text{O}_3$

Table 2: The FTIR peaks positions the  $(80-x)\text{TeO}_2-18\text{ZnO}-1\text{MgO}-1\text{Li}_2\text{O}-x\text{Er}_2\text{O}_3$  ( $1.0\text{mol}\% \leq x \leq 5.0\text{mol}\%$ )

Sample No	Composition (mol%)					IR bands ( $\text{cm}^{-1}$ )
	TeO <sub>2</sub>	ZnO	MgO	Li <sub>2</sub> O	Er <sub>2</sub> O <sub>3</sub>	
C1	79.5	18.0	1.0	1.0	0.5	458; 657; 755; 1110; 1636; 3417
C2	79.0	18.0	1.0	1.0	1.0	458; 660; 758; 1113; 1636; 3411
C3	78.5	18.0	1.0	1.0	1.5	458; 662; 758; 1113; 1636; 3411
C4	78.0	18.0	1.0	1.0	2.0	458; 665; 758; 1113; 1636; 3411
C5	77.5	18.0	1.0	1.0	2.5	458; 671; 758; 1113; 1636; 3411

## CONCLUSION

From the above discussions, the following conclusions may be drawn:

1. The addition of Er<sub>2</sub>O<sub>3</sub> content in the tellurite glasses results in the increase of T<sub>g</sub> from 315°C to 332°C, which implies in an increases in the rigidity of the network.
2. As the addition of Er<sub>2</sub>O<sub>3</sub> content is being increases from 0.5mol% to 2.5mol%, thermal stability of glass is increased from 97°C to 117°C.
3. The broad absorption band occurs around 660 $\text{cm}^{-1}$  and 760 $\text{cm}^{-1}$ , which can be ascribed to the stretching vibration of equatorial and axial Te–O bonds in the TeO<sub>4</sub> tpb unit and TeO<sub>3</sub> tp respectively.

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